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BiScO₃: Centrosymmetric BiMnO₃-type Oxide

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Bi- and Pb-containing perovskites with transition metals have received much attention in recent years due to the renewed great interest in multiferroic materials.¹⁻⁴ In multiferroic systems, two or all three of (anti)ferroelectricity, (anti)ferromagnetism, and ferroelasticity occur in the same phase. Such systems are rare in nature⁵ but have potentially wide technological applications.¹⁻⁴ The most attractive possible application is in multiple state memory elements.²

BiMnO3⁶⁻¹² and BiFeO3¹³ are the most studied multiferroic systems among BiMO₃ with magnetic M ions. BiMnO₃ is believed to crystallize in the noncentrosymmetric space group C2 to fulfill the requirement of ferroelectricity.^{10,11} This structural model is generally accepted for BiMnO₃ and related compounds, BiCrO₃¹⁴ and Bi₂MnNiO₆.³ Despite some efforts to understand the origin of ferroelectricity in BiMnO₃,^{6,7} it is not yet completely understood.^{8,12} On the other hand, little is known about simple Bi-containing compounds BiM'O₃ with nonmagnetic M' ions (M' = Al, Sc, Ga, and In). Recent theoretical studies of BiAlO3 and BiGaO3 predicted that they should be high-performance piezoelectrics and ferroelectrics with very large spontaneous polarization.¹⁵ Solid solutions including BiM'O₃ as the end members, for example, PbTiO₃-BiScO₃ and PbTiO₃-BiGaO₃,¹⁶ were extensively studied recently in the attempt to improve ferroelectric properties of PbTiO₃ and to reduce the amount of lead. In addition, BiM'O₃ can be model systems for theoretical understanding of the origin of ferroelectricity in multiferroic BiMO₃ because no additional complication of magnetic behavior exists in BiM'O₃.¹⁵

BiScO₃^{16,17} was prepared at 6 GPa in a belt-type high-pressure apparatus at 1413 K for 40 min. Neutron powder diffraction (NPD) data of BiScO₃ were collected at room temperature with a highresolution powder diffractometer (the JRR-3M rector in JAERI, Tokai, $\lambda = 1.8233(10)$ Å). Diffraction data were analyzed by the Rietveld method with RIETAN-2000. Indexing Bragg reflections in the X-ray powder diffraction and NPD of BiScO₃ revealed it to crystallize in the monoclinic system with lattice parameters of a =9.8899(5) Å, b = 5.8221(3) Å, c = 10.0462(5) Å, and $\beta =$ 108.300(3)°. These lattice parameters are similar to those of BiMnO₃ $(a = 9.5323 \text{ Å}, b = 5.6064 \text{ Å}, c = 9.8535 \text{ Å}, \text{ and } \beta = 110.667^{\circ})^{10}$ BiCrO₃,¹⁴ and Bi₂MnNiO₆,³ indicating that BiScO₃ should have a very similar structure. Reflection conditions derived from the indexed reflections of BiScO₃ were h + k = 2n for *hkl*, and h =2n and l = 2n for h0l, affording possible space groups Cc (No. 9) and C2/c (No. 15). The reflection conditions were checked and confirmed using electron diffraction. No reflections with l = 2n+1

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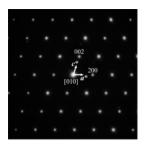


Figure 1. Electron diffraction pattern of BiScO₃ along the [010] zone axis.

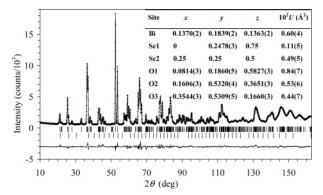


Figure 2. Observed (crosses), calculated (solid line), and difference patterns resulting from the Rietveld analysis of the neutron powder diffraction data for $BiScO_3$ at room temperature in space group C2/c. Bragg reflections are indicated by tick marks. The lower tick marks are given for the impurity phase, Sc_2O_3 (0.3 mass %). Final fractional coordinates and thermal parameters are listed.

were observed for h0l and 00l (Figure 1). These data unambiguously confirmed that the space group of BiScO₃ cannot be *C*2 (or *Cm* or *C*2/*m*) such as in BiMnO₃.¹⁰ No signal of second-harmonic generation was observed for BiScO₃, which shows it to be centrosymmetric. All these facts leave one possibility for BiScO₃, that is, space group *C*2/*c*.

Structure parameters of BiScO₃ were successfully refined from NPD in space group *C*2/*c* using the fractional coordinates of BiMnO₃¹⁰ transformed to space group *C*2/*c* as the initial ones. *R* factors were $R_{wp} = 5.38\%$ (S = 2.07), $R_p = 3.98\%$, $R_B = 1.70\%$, and $R_F = 0.83\%$. Thermal parameters for all the sites were typical and positive, and estimated standard deviations (esd's) of fractional atomic coordinates and thermal parameters were small (for example, $3-5 \times 10^{-4}$ for oxygen coordinates and 7×10^{-2} for oxygen thermal parameters; Figure 2). Figure 2 displays observed, calculated, and difference NPD patterns for BiScO₃.

The bond valence sum calculations resulted in the values of 2.89 for Bi, 2.91 for Sc1, and 2.92 for Sc2, confirming the formal

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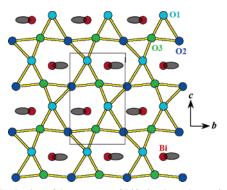


Figure 3. Projection of the structure of $BiScO_3$ along the *a* axis. One layer of O atoms (with bonds) and Bi atoms (with lone electron pair) is shown. Thin lines give a unit cell.

oxidation state +3 for Bi and Sc ions. Three Bi–O distances were short (2.15–2.25 Å), and other Bi–O distances were long (2.55–3.26 Å). Therefore, the lone electron pair of a Bi³⁺ ion is active in BiScO₃ (Figure 3), resulting in the antiparallel displacement of Bi³⁺ ions in the *b* direction from the centroids of their O coordination environments in the same manner as in BiMnO₃ with the *C*2 model.

We also tried to apply the *C*2 model as in the case of BiMnO₃, BiCrO₃, and Bi₂NiMnO₆ using the atomic coordinates of BiMnO₃¹⁰ as the initial ones in the structure analysis of BiScO₃. The *R* factors in the *C*2 model ($R_{wp} = 5.14\%$ (S = 1.99), $R_p = 3.82\%$, $R_B = 1.50\%$, and $R_F = 0.72\%$) were slightly smaller than those of the *C*2/*c* model because of the larger number of the refined structural parameters (39 vs 19). However, thermal parameters of two O atoms were negative. The esd's of fractional atomic coordinates (for example, $8-24 \times 10^{-4}$ for O) and thermal parameters (for example, 0.2-0.3 for O) of all the atoms were about 1 order of magnitude larger than one may expect from NPD data.

The esd's of fractional atomic coordinates and thermal parameters in the *Cc* model (the number of refined structural parameters was 38) were almost the same as those of the *C*2 model with slightly smaller *R* factors in the *Cc* model: $R_{wp} = 5.02\%$ (*S* = 1.94), $R_p =$ 3.70%, $R_B = 1.33\%$, and $R_F = 0.56\%$. Two O atoms also had negative thermal parameters in the *Cc* model. The negative thermal parameters for O atoms and large esd's for all the sites obtained from NPD data are good indicators that the symmetry of BiScO₃ is higher than *C*2 or *Cc* and a center of symmetry is missed.¹⁸ It is known that disregarding a center of symmetry results in errors in the atomic coordinates that are large compared with normal esd's.¹⁸

Checking the available structural data for BiMnO₃ and related compounds revealed the same problem, which was systematically ignored, with the C2 model. In the first paper reporting the crystal structure of BiMnO₃ from NPD at room temperature, esd's for O coordinates were $1-5 \times 10^{-3}$.¹⁰ In ref 11, using NPD at 20 K, the thermal parameters for all the sites were fixed, and esd's for O coordinates were $2-9 \times 10^{-3}$. These esd's of O coordinates are very large for NPD data. In the structural analysis of Bi₂NiMnO₆ from high-quality synchrotron XRD, the authors had to refine the common thermal parameters for all Bi sites, all Ni/Mn sites, and all O sites.³ The esd's of Bi coordinates were $1-12 \times 10^{-3}$, which are extremely large for heavy Bi atoms.

All of the above facts and structural information obtained for BiScO₃ in this work indicate that the structural model for BiMnO₃ and related compounds should be revised. There is good indication that the atomic arrangements in the bulk BiMnO3 are very close to centrosymmetric, if not centrosymmetric at all. Therefore, it seems that the structural distortion cannot be the origin of ferroelectricity in bulk BiMnO₃ as generally accepted. In ref 6, it was noted that noncentrosymmetric distortion in BiMnO₃ is most unusual for the d^4 Mn³⁺ Jahn–Teller ion. In refs 7 and 12, it was suggested that ferroelectricity in BiMnO₃ may originate from an electronic phase transition (electronic ferroelectricity)¹⁹ in contrast to the conventional displacive ferroelectricity caused by a lattice distortion. In ref 20, a very small polarization was obtained for BiMnO₃ in the C2 model from first-principles calculations. BiScO₃ may serve as a model system to understand a role of Mn³⁺ ions in ferroelectricity of BiMnO₃. We hope that the results of this work will stimulate further research activities in studying the intriguing question about the origin of ferroelectricity in BiMnO₃.

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Supporting Information Available: Details of experiments; selected interatomic distances in the C2/c model; observed, calculated, and difference NPD patterns and structure parameters for BiScO₃ refined in space groups C2 and Cc (PDF). This material is available free of charge via the Internet at http://pubs.acs.org.

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